10-047-009 (REV. 6/63)

CR-72243

DIVISION SNAP-8

TM 4923:65-4-357

DATE 3 December 1965

w.o. 0740=05-2003

TECHNICAL MEMORANDUM

AUTHOR(S):

H. E. Bleil

TITLE:

DIMENSIONAL CHANGE OF STELLITE 6B DURING PHASE TRANSFORMATION FROM FACE-CENTERED CUBIC TO HEXAGONAL CLOSE-PACKED CRYSTAL STRUCTURE

ABSTRACT

The linear dimensional contraction of Stellite 6B due to transformation from the face-centered cubic to the hexagonal close-packed crystal structure was determined on 18 parts using 9l individual measurements. The average contraction and the standard deviation of the 9l individual measurements were determined to be 0.092% and 0.042%, respectively.

N67	-32575	
	CCESSION NUMBER)	(THRU)
ē	\mathscr{G}	1
	(PAGÉS)	(CODE)
½ _ <u>U</u>	7-72243	17
(NASA CI	OR TMX OR AD NUMBER)	(CATEGORY)

APPROVED:

DEPARTMENT HEAD

P. J. Wood

NOTE: This document is considered preliminary and is subject to revision as analysis progresses and additional data are acquired. The general reader may encounter internal reference not available to him.

INTRODUCTION

Stellite 6B* is a high-temperature cobalt base alloy which exhibits excellent mercury erosion resistance and good mercury corrosion resistance. Therefore, it was selected and employed as the reference SNAP-8 turbine rotor and nozzle material. It was subsequently determined that at the turbine operating temperatures (700-1238°F) this alloy, in the forms employed in the turbine, transforms from a face centered cubic (FCC) to a hexagonal close packed (HCP) crystal structure. This transformation was produced in the laboratory by thermally treating Stellite 6B for 4 hours at 1650°F followed by 48 hours at 1250°F. The change in structure is accompanied by a volumetric contraction of the material. Therefore, machined parts of Stellite 6B in the FCC condition undergo a reduction in linear dimensions when transformed to the HCP condition. To determine the extent of this contraction and its effect on drawing tolerance requirements, various samples and actual parts were dimensionally inspected before and after the transformation and the data were analyzed statistically.

TEST PROCEDURE AND RESULTS

Various laboratory samples prepared from forgings and bar stock, and thirty-seven actual TA parts were transformed using the laboratory thermal treatment (see Table I). Ninety-one individual linear dimensional measurements were made before and after transformation on eighteen of the pieces. The number of dimensions obtained on each piece was dependent on the configuration at the part being inspected. The laboratory samples were primarily thin, narrow rectangular specimens and only the longest dimension was inspected in two locations. On actual circular parts, however, several dimensions at different circumferential locations around the periphery of the pieces were obtained. The calculated average contraction and standard deviation were 0.092 and 0.042%, respectively (see Table II and Figure 1). Based on these data and on the assumption that a normal distribution exists, there is a 99% probability that 99% of the measured contractions due to the crystallographic

^{* 30}Cr - 4.5W - 1.1C - 3 max Ni - 2 max Si - 3 max Fe - 2 max Mn - 1.5 max Mo

transformation of Stellite 6B will fall between 0% and 0.223% (see Figure 2).

Based on a plot of the individual data points (Figure 1) the assumption of normal distribution is essentially true, but the curve stops at 0% contraction because of the physical impossibility of growth occurring during the transformation from FCC to HCP. This impossibility exists because the HCP structure forms from rearrangement of the same number of atoms as the original FCC structure but in a more dense population or smaller interatomic spacing pattern. The more densely packed atoms cannot produce a larger volume of metal. The data (Table II) indicate that a significantly greater variation exists between specimens than within individual specimens. This is demonstrated by the standard deviations of 0.030% and 0.075% for variability within and between specimens respectively, and by the 6.4 F ratio obtained. An F ratio this large would be obtained less than one time in a thousand if the specimens were all being randomly selected from a homogeneous population. The large variation in contraction, both within and between specimens, remains unexplained at present. However, it is thought that anisotropy and possibly residual stress patterns may influence the dimensional changes which occur. literature indicates investigators have found that the linear contraction of pure cobalt due to transformation lies between 0.08 and 0.12%. This relatively wide variation also has not been explained but it has been theorized that anisotropy and impurities may influence the contraction.

CONCLUSIONS

- 1. The crystallographic transformation of Stellite 6B from a FCC to a HCP crystal structure is accompanied by a volumetric contraction of the alloy.
- 2. The mean linear contraction and standard deviation on data from 18 specimens are 0.092% and 0.042%, respectively.
- 3. There is a significantly greater variation in contraction between specimens than within individual specimens.
- 4. The actual amount of contraction which will occur on a specific part cannot be predicted with enough accuracy to permit the transformation to be performed on a finish machined part for the turbine unless wider than presently desirable tolerances are allowed.

RECOMMENDATIONS

- 1. Stellite 6B parts should be transformed to the HCP condition prior to the final machining operation whenever possible to eliminate the effect of contraction variability on part tolerance.
- 2. The anticipated range of linear contraction, 0.000% to 0.223%, should be examined in relation to specific parts which have been finish machined prior to the transformation to the HCP condition to determine if the anticipated contraction is acceptable.
- 3. Further dimensional contraction data in three axes should be obtained on Stellite 6B parts which are to be transformed. This will improve the statistical understanding of the phenomena and could indicate if anisotropy contributes to the variation in contraction.

TABLE I

SPECIMEN DISCRIPTIONS

ed (in.)*	Minimum	1,0028	1.7473	1.7474	1.7474	1.7472	6747.	1.7476	1.7475	1.7475									1.7476
Part Dimension Measured (in.)*	ЗI	П	Н	П	Н	П	П	Т	L	П	2.1294	2.1180	2.2098	2.2090	2.3110	2.3.43	2.1982	2.0563	Т
Part Dimens	Maximum	5.4504	6.2052	6.2063	6.2060	6.4070	6.4071	6.4064	6.2053	6.4067									5.5848
+41																			
		Rotor	Nozzle Diaphragm Assembly	Nozzle Diaphragm Assembly	Specimen	Specimen	Specimen	Specimen	Specimen	Specimen	Specimen	Specimen	Fourth-Stage Diaphragm						
	Name	First-Stage Rotor	Nozzle Diap	Nozzle Diapl	Nozzle Diap	Laboratory Specimen	Laboratory Specimen	Laboratory	Laboratory Specimen	Laboratory Specimen	Laboratory	Laboratory Specimen	Laboratory Specimen	Fourth-Stag					
	No.																		
	Specimen No.	⊣	a	2	7	7	9	7	∞	6	10	11	12	13	1,4	15	16	17	18

^{*} All dimensions were read to the nearest 0.0001 inch. The estimated error in reading, therefore, is \(^+\).005% for the smallest dimension, 1.0028, and \(^+\).0008% for the largest dimension, 6.4071.

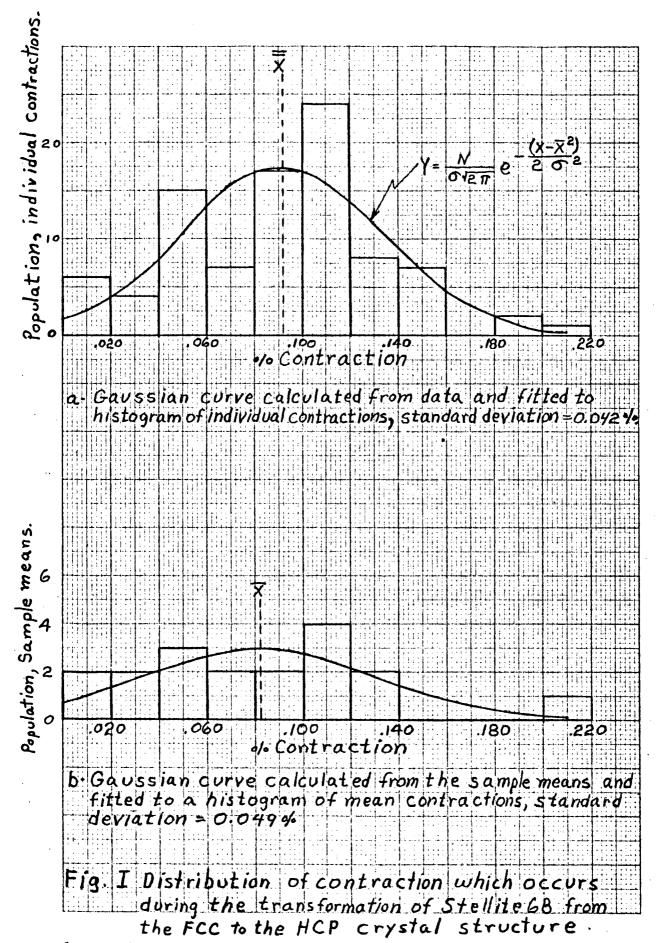
TABLE II

ANALYSIS OF CONTRACTION OF STELLITE 6B IN TRANSFORMING FROM FCC TO HCP CRYSDAL STRUCTURE

															Totals	8.3%	6.771656	91	.871406	.936272	.064861	1.481
13	.015															.015	.000225	т	.000225	.000225	0	.015
17	840.			** serve			ş gerdinke w					THE STATE OF	*****	-		840.	.002500	ч	.002304	.002304 .000225	0	840.
16	.210	. Au														.210	.044100	H	.044100	001440.	, O	.210
15	.118	.027						••				Theorem a gho	and year and			.145	.021025	N.	.010512	.014653	041400.	.072
14	.017	.052	ann - A-special had								alan ar no t		-			690.	.004761	ณ	.002380	.002993	.000612	.034
13	.059	.05h			un - um a.	4.4 40.40.	enement were									it.	.012769	N	,006384	.003065 .006397 .002993	.000012	.057
21	610.	.052				wie west o						•				۲70.	.005041	a	.002520	.003065	· 000544	.036
Ħ	.048	.053						,								101.	.010201	a	.005101	.005113	.000012	.050
01	.683	8	·			·		april p								.023	.000529 .010201	a	±92000.	.083902 .000529	.000264	510.
6	80.	.039	. 500	.195			afri sear		****						•	.500	.250000	.	.062500 .000264	.083902	.0214ce	ध्यः
80	.134	.145	.148	121.	4	.149	811.	.149								1.085	.511225 1.177225	00	.147153	148693	.001540	.136
<u> </u>	.081	690.	650.	.058	120	.109	011.	.109								.715	.522115.	ω	.063903	.068429 .148693	.004526	869
9	911.	.101	.125	.095	.125	.109	.133	.109								.913	.833569	8	.104196	.105383	.001187	† 11:
5.	.106	211.	.070	86.	.086	.109	%	.109						-		942.	.556516	89	₹9690•		.002582	.093
4	.100	1	910.	.133	160.	.143	86	.143				-				.841	.707281	80		7,1001.	192110.	.105
٤	100	.120	оп. -	911.	.097	151	.097	.143								.80	.846400	80	.105800	.107992	.002192	.115
ผ	011.	1 80.	901.	.129	901.	160.	ij	760.			•		•			548.	.708964	80	.088620	.089836	.001216	.105
н.	.055	.055	.057	.057	790.	.082	.077	.082	₹ 8	.05	8	980.	.100	.120	8	1.039	(\$ x)2 1.079521 .708964	79	024790.	£ X ²080335 .089836107992 1.100177 1.072146	.012865	.065
Specimen No.						· • •			πο	sett	n d to	100 g	į,		•• •••	×	(x x) ₂	E	008901. 058820 .0470. A ⁵ (×≯)	× ×	εχ ^ε -(εχ) ² .	k

F Ratio	6.4* X = .0924 X = .0924	
Standard Deviation	. 0304 . 0304 . 040	
Mean Square	.005692 .000889 .001796	
Degrees of Freedom	258	
Sum of Squares of Deviation	.0%765 .064861 .161626	
Source of Variation	Between Specimens Within Specimens Total	

* P (F = 6.4 d1 = 17, df2=73) < .001; Variation between specimens is concluded to be significantly greater than variation within specimens.



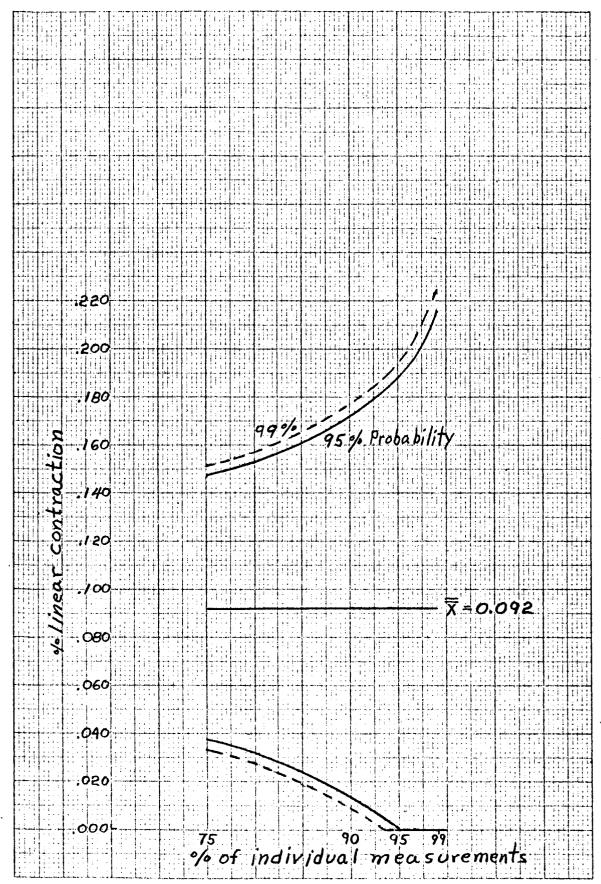


Fig. 2. Probability of individual contractions falling within specific tolerances.